THE S-D TRANSITION AND HIGH-PRESSURE PHASE STABILITY OF TRANSITION METALS

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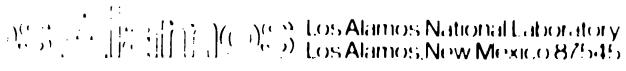
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THE S-D TRANSITION AND HIGH-PRESSURE PHASE STABILITY OF TRANSITION METALS*

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ABSTRACT

The recent discovery of a solid-solid phase change in shock-compressed Mo and the theoretical interpretation suggest valence d-electron density as a major influence on structural stability. The relationship of this experimental result to the transition metal structures and alloy phase diagrams will be discussed. Specific predictions will be presented for the locations and slopes of transition metal and transition metal alloy phase boundaries.

KEY WORDS

Transition metals, phase transition

A solid-solid phase transformation has recently been reported for Mo at high shock pressures (Hixson et al., 1989). This phase transition, detected by an abrupt change in elastic wave velocity of shock compressed Mo, has been interpreted in terms of the pressure-induced shift of conduction electrons from s-like states to d-like states (McMahan, 1986). Previous theoretical (McMahan, et al., 1981) and experimental (Brown et al., 1985) work on Lanthanum have shown that the gradual shift of s-like to d-like electrons can have large effects on the equation of state, particularly in the derivatives, Gruneisen's gamma and the bulk modulus

Skriver has shown through detailed LMTO calculations that the phase stability of the transition metals is dependent on the conduction d-electron density (Skriver, 1985). He show: that the stability of the bcc phase for group-V and VI transition metals at low temperature and pressure is dependent on having between 2.5 and 5 d-like electrons per ion. The group-VI metals Cr. Mo, and W need roughly 0.5 more d-electron per ion to cause stabilization of a close-packed phase, while the group-IV metals need only a small increase in d-electron occupancy to cause stabilization of the box phase.

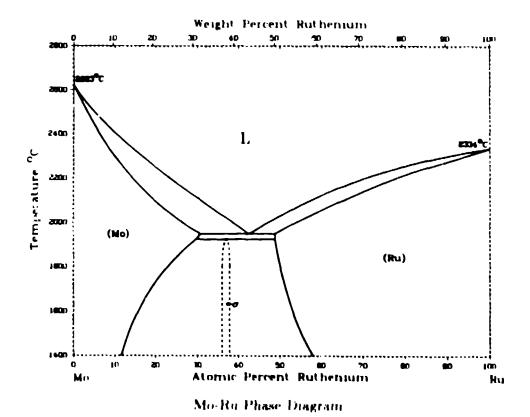
In addition to pressure or increase in atomic number, another way to increase the delectron occupation is by alloying with a higher-z metal. A phase diagram of the Mo-Ru system is shown in Fig. 1. The maximum ruthenium concentration in the bic phase alloy is 30 atomic percent in the Mo-rich component of the entertic. The phase diagrams for Mo-alloys with To-Ru, Rh, and Pd. as well as those for W-alloys with Re-Os-fr and Pt are all similar (Massalski, 1986). We show the maximum solute concentrations in the bic phase and the average number of electrons per ion in Table 1. From this table we see that adding roughly 0.5 electron per ion to the Mo-system and 0.4 electron per ion to the W-system will destabilize the bic lattice. These numbers are roughly consistent with Skriver's calculations.

Another recent calculation confirming the dielectron occupation dependence of ; has stability has been presented for Re (Watson et al., 1998). APW calculations how that as pressure is increased on Re sausing the shift from a like to dilike confuction electrons, the help phase temains stable in contrast to the auggestion that

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there may be a high-pressure bcc phase. On the basis of the unhybridized band calculations one expects pressure to drive Re away from the region of stability for bcc.



At the high temperatures and pressures in a strong shock, phase stability calculations must include all the terms in the Gibbs free energy. In particular the phonon entropy term which stabilizes the bcc structure at high temperature is important (Animalu, 1967). The electron entropy term will be generally smaller so long as $I \geq 0.1$ Ty, the Fermi temperature. Using a rigid band model, this term which is proportional to the density of states at the Fermi surface favors the bcc structure except for systems with six valence electrons per atom (Morozzi, 1978). The other term in the free energy differences, $P\Delta V$, will also become important at sufficiently high pressures. Another complication in the very high pressure phase diagram is the hybridization of the s and d bands at pressures high enough that all valence electrons behave a like. We expect again to stabilize the bcc lattice (Ho iver et al., 1972).

Table I

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Alloy	Atomic "? Solute	Conduction Electrons Per Ion
Mo-To	50	6.5
Mo-Ru	30	6 G
Mo-Rb	20	6.6
Mo-Pd	7	6.3
W-Re	37	6.4
W-Os	19	6.4
W-Ir	10	6.3
W-Pt	5	6 2

We have shown that experiment and theory consistently suggest that d-electron occupation plays a dominant role in crystal structure stability and thermodynamics of transition metals. These observations suggest several more theoretical and experimental tests of the underlying physics. We know, for example that the hop-box boundary for the group-IV elements, Ti, Zr, and Hf has a negative dP/dT. This suggests that pressure stabilizes the box physic along an isotherm, consistent with the increase in d-band occupancy. However, the other terms in the Gibbs free energy should be evaluated to show whether this phenomenon is really d-electron different alloying and pressure appear to have the same effect on structural stability by alloying Mo with Ru or W with Re, for example, one should force the box-hop phase boundary to lower pressures, where it may be accessible to techniques other than strong shock loading

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